## UDC: 669.181.4:669.784 Grishin A.M., Ivashchenko V.P., Nadtochij A.A., Shcheglova I.S. THERMODYNAMIC REGULARITIES OF OBTAINING SPONGY LIGATURES WITH LOW CARBON CONTENT

A thermodynamic analysis was carried out on the subject of metal oxides and ferrites reduction both with chromium carbide Cr<sub>7</sub>C<sub>3</sub> individually and in combination with a hydrogen stream to obtain low-carbon chromium-based ligatures. It was shown the possibility of organizing such a process in the temperature range excluding charge melting. The obtained data confirmed that the proposed method can produce sponge ligatures based on chromium of complex composition.

Keywords: chromium carbide, sponge ligature, thermodynamic analysis, carbon-thermal and complex reduction, oxide, ferrite.

Виконано термодинамічний аналіз процесів відновлення оксидів металів і феритів карбідом хрому Cr<sub>7</sub>C<sub>3</sub> самостійно і в комбінації з потоком водню стосовно отримання лігатури на основі хрому з низьким вмістом вуглецю. Показана можливість організації такого процесу в області температур, що виключають плавлення шихти. Отримані дані дозволяють зробити висновок, що запропонованим способом можна отримати губчасті лігатури на основі хрому комплексного складу.

Ключові слова: карбід хрому, губчаста лигатура, термодинамічний аналіз, вуглецево-термічне та комплексне відновлення, оксид, ферит.

Sponge and powder ligatures have a unique complex of technological properties, which makes it possible to use them for special steel grades and alloys smelting, as well as in powder metallurgy. The use of spongy and powder ligatures can significantly reduce the cost of basic and auxiliary materials, energy resources and to extract alloying elements from ores to a greater extent.

The use of complex sponge ligatures also makes it possible to significantly increase operating properties of the products obtained [1, 2]. However, up to the present time, the problem of obtaining of complex composition sponge ligatures with low carbon content by the solid-phase reduction method has been insuffi-

 $NiO + CO = Ni + CO_2$  $+ 1/3Cr_7C_3 + CO_2 = 7/3Cr + 2CO$  $1/3 Cr_7C_3 + NiO = Ni + CO + 7/3Cr$ 

The link of carbide gasification, in accordance with the principle of the sequence of transformations, is realized as:  $Cr_7C_3 \rightarrow Cr_{23}C_6 \rightarrow Cr_{sat-d C}$ . The physicochemical model of Cr<sub>7</sub>C<sub>3</sub> gasification can be carried out in a stepwise or zonal scheme depending on the diffusion and chemical parameters. Under the conditions providing the zonal gasification regime, the following transformations take place simultaneously:  $Cr_7C_3 \rightarrow Cr_{23}C_6$  and  $Cr_{23}C_6 \rightarrow Cr$ , which are spatially separated in the volume of the carbide particle.  $1/3 \text{ Cr}_7\text{C}_3 + \text{NiO} = \text{Ni} + \text{CO} + 7/3 \text{Cr}$ 

1/3 C 1/6Cr<sub>23</sub>C<sub>6</sub>+NiO=23/6Cr+Ni+CO

As can be seen in Figure 1, at temperatures above ~ 1073 K, the gasification process is thermodynamically more preferable for reactions 4 and 6. Proceeding from this, the gasification of the carbide Cr<sub>7</sub>C<sub>3</sub> was further examined by reaction 4.

Thermodynamic parameters that determine the possibility of reducing the oxide (or more complex compound) at a constant pressure are the temperature and composition of the gas phase. The presented such a process was analyzed for the reduction of nickel oxide by trigonal chromium carbide. A purely carbothermic as well as complex reduction in the CO flux can be described by a set of reactions:

ciently studied. In some publications [1, 3] the authors

propose a solution with a limited O/C ratio, which sig-

modynamic bases for obtaining a complex ligature by

the solid-phase reduction of oxides with the participa-

tion of chromium carbide. The proposed method for

The aim of this work is the development of ther-

nificantly narrows the composition of ligatures.

(1)

(2)(3)

( 1)

Indirect confirmation of the zonal flow of the process is obtained by the composition of the gaseous phase,  $(Cr_7C_3 \rightarrow Cr_{23}C_6)$  + formed at the stage of A high concentration of CO2 initiates NiO. simultaneously the second stage of gasification of  $Cr_{23}C_6 \rightarrow Cr$  and further reduction of nickel oxide. Possible variants of carbide participation during the reduction of nickel oxide are represented by the reactions:

$$r_7C_3 + 7NO = Ni + CO + 7/3Ci$$
(4)  

$$r_7C_3 + 27/69NiO = 7/69Cr_{23}C_6 + 27/69Ni + 27/69CO$$
(5)  

$$r_7C_3 + NiO - 23/6Cr + Ni + CO$$
(6)

system (reactions 1 - 3) is formed by four components (K) and contains five phases (F). It has, according to the rule of phases: V = K + 2-F, one degree of freedom (V - variance). At a fixed pressure of carboncontaining gases ( $P_{CO} + P_{CO2} = \alpha$ ), equilibrium occurs under conditions of a single, completely defined temperature (Teq.). It can be determined on the basis of the equilibrium constants of the reactions (1) and (2). The latter can be represented in the form:

obtaining the master alloy consists of the complex reduction of oxides mixture by the previously obtained chromium carbide Cr<sub>7</sub>C<sub>3</sub>. Initially, the possibility of

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Representing the equilibrium constant of the link  $K_1 = \frac{(\alpha - P_{\rm CO})}{P_{\rm CO}}$ (1) in the form

we obtain the ratio 
$$P_{CO} = \frac{\alpha}{(1+K_1)}$$

$$\alpha = \exp\left(a_0 + \frac{a_1}{T}\right) \cdot \exp\left(B_0 + \frac{B_1}{T}\right) \cdot \left[1 + \exp\left(a_0 + \frac{a_1}{T}\right)\right]$$

For some oxides, the start temperature of reduction was calculated for different values of a. The results are shown in Table 1.

The equilibrium composition of the gas phase also determines the thermodynamics of the process. The equations, obtained earlier, allow us to determine the composition of the gas phase in the reaction (1) at temperatures  $T \leq Teq.$ :

$$P_{\rm CO} = \alpha/(1 + K_1)$$
 and  $P_{\rm CO_2} = \alpha - P_{\rm CO}$ .

For the zonal or stepwise mode of the gasification process, the composition of the gas phase will be dif-

Fig.1. Temperature dependence of free energy change  $(\Delta G^0)$  in reactions: 1.  $1/3 Cr_7C_3 + NiO = Ni + CO + 7/3Cr;$ 2. 1/3Cr7C3+27/69NiO=7/69Cr23C6+27/69Ni+27/69CO; 3. 1/6Cr<sub>23</sub>C<sub>6</sub>+NiO=23/6Cr+Ni+CO.

Equating the two expressions of  $P_{CO}$ , we get the equation for calculating the temperature of the beginning of the carbothermic reduction of NiO (Teq.):

$$\alpha = \mathbf{K}_1 \cdot \mathbf{K}_2 \cdot (1 + \mathbf{K}_1) \tag{7}$$

The equation presented earlier [4,5] allows to determine the value of Teq. for various values of  $\alpha$ : **۲** 

$$\left[ -\frac{1}{2} \right] , \qquad (8)$$

ferent, which will affect the thermodynamics of the nickel oxide reduction by Cr<sub>7</sub>C<sub>3</sub> carbide. The composition of the gas phase in the zonal gasification regime can be approximately calculated from the reaction:

$$\frac{1}{3} \operatorname{Cr}_{7} \operatorname{C}_{3} + \operatorname{CO}_{2} = \frac{7}{3} \operatorname{Cr} + 2\operatorname{CO}$$
  
K<sub>2</sub> =  $\frac{\operatorname{P}_{CO}^{2}}{\operatorname{P}_{CO_{2}}} = \frac{\operatorname{X}^{2}}{\alpha - \operatorname{X}}$ , hence  $\operatorname{X}^{2} + \operatorname{K}_{2} \cdot \operatorname{X} - \operatorname{K}_{2} \cdot \alpha = 0$ 

, where  $\alpha = P_{CO} + P_{CO2}$ 

The data obtained on the composition of the gas phase for the Ni-Cr-O-C system are shown in Fig. 2.

						Table 1		nperature	S OI OXIUE	reduction
α	0,15	0,25	0,35	0,45	0,55	0,65	0,75	0,85	0,95	1,0
NiO	727.8	745.5	757.8	767.2	774.8	781.3	786.9	792.0	796.5	798.6
Fe <sub>2</sub> O <sub>3</sub>	459.9	466.5	470.9	474.3	477.1	479.4	481.4	483.1	484.7	485.4
Fe <sub>3</sub> O <sub>4</sub>	1184.9	1209.9	1227.2	1240.2	1250.8	1259.9	1267.6	1274.5	1280.7	1283.5
FeO	1277.1	1313.2	1338.1	1357.2	1372.8	1386.2	1397.8	1417.4	1417.4	1421.7
MoO <sub>2</sub>	1273.6	1304.1	1324.9	1340.7	1353.8	1364.8	1374.5	1383.0	1390.7	1394.0
WO <sub>3</sub>	1055.2	1085.5	1106.4	1122.6	1135.9	1147.2	1156.9	1165.7	1173.5	1177.1
WO <sub>2</sub>	1273.6	1304.1	1324.9	1340.7	1353.8	1364.8	1374.5	1383.0	1390.7	1394.2
Fe <sub>2</sub> MnO <sub>4</sub>	1413.8	1449.3	1473.7	1492.6	1507.9	1520.9	1532.3	1542.4	1551.4	1555.6

Start temperatures of oxide reduction Tabla 1

According to the proposed method, the equilibrium composition of the gas phase is calculated for the reduction of iron oxides by Cr<sub>7</sub>C<sub>3</sub> carbide, the results are shown in Fig. 3. The reduction proceeds successively  $Fe_3O_4 \rightarrow FeO \rightarrow Fe$ , but the process regime is more likely to be stepwise, however, this requires additional analysis.

Table 1 presents the results of calculating Teq. of iron oxides reduction with chromium carbide, which correlate with figure 3. The sequence and mechanism of solid product formation in this process are insufficiently investigated.



Fig. 2. Equilibrium in the Ni-Cr-C-O system 1. NiO + CO = Ni + CO<sub>2</sub> 2.3. 1/3 Cr<sub>7</sub>C<sub>3</sub> + CO<sub>2</sub> = 7/3 Cr + 2CO at  $\alpha$  = 0.25 and 0.75, respectively

> Fig.3. Equilibrium in Fe - Cr - C - 0 system: 1, 2 - 1/3Cr<sub>7</sub>C<sub>3</sub> + CO<sub>2</sub> = 7/3Cr + 2CO, if  $\alpha = 0,25$  and 0,75; 3 - FeO + CO = Fe + CO<sub>2</sub>; 4 - Fe<sub>3</sub>O<sub>4</sub> + CO = 3FeO + CO<sub>2</sub>.

Along with the metal oxides, other oxygencontaining compounds may also participate in the reduction process, for example, the components of the rolling mill scale - ferrites:  $NiFe_2O_4$ ,  $CoFe_2O_4$ ,



 $Fe_2MnO_4$ , etc. Thermodynamic analysis has shown the possibility of their reduction by chromium carbides under given temperature conditions. The results are presented in Fig. 4.

Fig.4. Carbon-thermal reduction of ferrites by  $Cr_7C_3$ : 1, 2 – gasification of carbide with  $CO_2$  at  $\alpha$  = 0.25 and 0.75 respectively;

 $3 - NiFe_2O_4$ ;  $4 - CoFe_2O_4$ ;  $5 - Fe_2MnO_4$ .

However, some difficultly reducible oxides under the given temperature conditions cannot be thermodynamically reduced with chromium carbides. The introduction of such metals into the ligature is possible with their carbides, which also will increase the reduction capacity of the system. The participation of the carbide phase in the prosecond from the transformation of the carbide phase in the pro-

cess of carbothermic or complex reduction is realized through the gasification unit [6]. The possibility of using the reduction potential of carbides (introduced or formed) at given temperature limits can be estimated from the temperature of their gasification beginning. For a number of carbides, the obtained values of the gasification start temperature are shown in Table 2.

The results show that all this carbides, except  $Mn_7C_3$  and  $MnC_2$ , gasify at a given temperature range. To come over the thermodynamic prohibition, emerging for individual carbides, is possible with use of a stronger oxidizer - oxygen, or by the introduction of

this metal into the initial charge in other chemical form. The real picture of the carbide phase formation and participation in the process of carbothermic reduction of oxides is much more complicated due to the formation of complex carbides of variable composition, for example  $M_7C_3$ , where M is Cr, Fe, Ni. To assess the physicochemical patterns of their participation in the reduction process is possible with thermodynamic modeling methods using special programs (for example – Fact Sage). Introduction of hydrogen to the system, i.e. transition to a complex reduction involving chromium carbides, significantly changes the thermodynamic picture of the process. In parallel with reactions 1, 2, the following reactions are triggered:

$$NiO+H_2=Ni+H_2O$$
(9)

$$1/3Cr_7C_3 + H_2O = 7/3Cr + H_2 + CO$$
 (10)

$$H_2 + CO_2 = CO + H_2O \tag{11}$$

Element	Carbida	T, K at $\Delta G^{\circ} = 0$			
	Carbide	CO <sub>2</sub>	H <sub>2</sub> O		
Fe	Fe₃C	958.885	926.532		
	Cr <sub>3</sub> C <sub>2</sub>	1093.57	1093.17		
Cr	Cr <sub>7</sub> C <sub>3</sub>	1352.24	1413.26		
	$Cr_{23}C_6$	1384.68	1454.67		
Ni	Ni₃C	634.171	521.616		
	Mn <sub>7</sub> C <sub>3</sub>	1587.96	1682.09		
	Mn <sub>3</sub> C	-	-		
Mn	Mn <sub>15</sub> C <sub>4</sub>	-	-		
	Mn <sub>23</sub> C <sub>6</sub>	1228.42	1252.19		
Мо	MoC	1042.12	1030.4		
	Mo <sub>3</sub> C <sub>2</sub>	805.531	727.224		
	Mo <sub>2</sub> C	1329.61	1378.23		
14/	WC	1227.16	1248.23		
VV	W <sub>2</sub> C	1119.34	1125.3		
N/	VC	1418.73	1485.22		
V	V <sub>2</sub> C	1513.45	1595.17		

Table 2- The start te	emperatures of	t carbides	aasitication
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The latter, together with the reaction (2), makes it possible to describe adequately the unit of steam gasification of  $Cr_7C_3$ . The peculiarity of this link is characterized by the fact that the formation of hydrogen is accompanied by the appearance of a mixture of CO-CO<sub>2</sub>, the ratio of which is determined by the temperature and the proportion of carbon- and hydrogen-containing gases in the reaction mixture. Reac-

tion 9 can be replaced by reactions 3 and 11. The temperature of the complex reduction beginning corresponds to the onset of steam gasification of the carbide. Calculation of the equilibrium composition of the gas phase was carried out according to the procedure described earlier [7]. Figure 5 shows the results of calculating the equilibrium in the Cr-Ni-C-H-O system.



Fig.5. Equilibrium in Cr - Ni - O - C - H system at: a)  $\alpha = 0.75, \beta = 0.25; \beta = 0.25; \beta = 0.75$ 

Hydrogen positively influences on the thermodynamic conditions of the process, reducing the temperature. Simultaneously, partial replacement of carbon in carbide phase by hydrogen promotes the decarburization of the product, and also allows expanding the composition of the initial charge. The water vapor formed in this case thermodynamically favors the gasification of carbide [8], being a stronger oxidant. However, a detailed physical-chemical analysis is needed to determine the contribution of hydrogen and carbide carbon in the reduction process in order to minimize the residual carbon in the system.

All the above thermodynamic features are also retained in the complex reduction of ferrites. Figure 6 shows the temperature dependence of the equilibrium composition of the gas phase of the complex reduction of  $Fe_2MnO_4$ . Increasing the proportion of hydrogen significantly reduces the start temperature of reduction.



Fig.6. Complex  $Fe_2MnO_4$  reduction by  $Cr_7C_3$  carbide: 1, 2 - carbide gasification with  $H_2O$ ; 3. 4 – ferrite reduction: 1, 3 -  $\alpha$  = 0.25,  $\beta$  = 0.75; 2, 4 -  $\alpha = 0.75$ ,  $\beta = 0.25$ .

The results in Fig. 6 relate to the reaction of the complete reduction of manganese ferrite with chromium carbide:

However, other process schemes are possible, leading to the formation of a more complex solid product. The determining parameters in this are: temperature, the ratio O/C and C/H. We have analyzed alternative

 $Fe_2MnO_4 + 1/3Cr_7C_3 + 3H_2 = 2Fe + 7/3Cr + Mn + CO + 3H_2O$ (12)options for both carbon-thermal and complex reduction of manganese ferrite. As possible variants, depending on the given conditions, reactions were chosen (per 1 mole of  $O_2$ ):

$1/4Fe_2MnO_4 + CO = 1/2Fe + 1/4Mn + CO_2$	(13)
1/2Fe <sub>2</sub> MnO <sub>4</sub> +2/3Cr <sub>7</sub> C <sub>3</sub> =Fe+14/3Cr+1/2Mn+2CO	(14)
1/2Fe <sub>2</sub> MnO <sub>4</sub> + 49/27Cr <sub>7</sub> C <sub>3</sub> = Fe + 343/621Cr <sub>23</sub> C <sub>6</sub> + 1/46Mn <sub>23</sub> C <sub>6</sub> + 2CO	(15)
1/2Fe <sub>2</sub> MnO <sub>4</sub> +46/27Cr <sub>7</sub> C <sub>3</sub> =Fe+14/27Cr <sub>23</sub> C <sub>6</sub> +1/2Mn+2CO	(16)
1/2Fe <sub>2</sub> MnO <sub>4</sub> +49/69Cr <sub>7</sub> C <sub>3</sub> =Fe+343/69Cr+1/46Mn <sub>23</sub> C <sub>6</sub> +2CO	(17)
1/2Fe <sub>2</sub> MnO <sub>4</sub> + 3/2CO+ 1/2H <sub>2</sub> =Fe + 1/2Mn + 3/2CO <sub>2</sub> + 1/2H <sub>2</sub> O	(18)
1/2Fe <sub>2</sub> MnO <sub>4</sub> + 1/6Cr <sub>7</sub> C <sub>3</sub> +3/2H <sub>2</sub> = Fe + 7/6Cr + 1/2Mn +1/2CO+3/2H <sub>2</sub> O	(19)
1/2Fe <sub>2</sub> MnO <sub>4</sub> +26/27Cr <sub>7</sub> C <sub>3</sub> +H <sub>2</sub> = Fe + 182/621Cr <sub>23</sub> C <sub>6</sub> + 1/46Mn <sub>23</sub> C <sub>6</sub> +CO+H <sub>2</sub> O	(20)
1/2Fe <sub>2</sub> MnO <sub>4</sub> +23/27Cr <sub>7</sub> C <sub>3</sub> +H <sub>2</sub> = Fe + 7/27Cr <sub>23</sub> C <sub>6</sub> + 1/2Mn +CO+H <sub>2</sub> O	(21)
1/2Fe <sub>2</sub> MnO <sub>4</sub> +26/69Cr <sub>7</sub> C <sub>3</sub> +H <sub>2</sub> = Fe + 182/69Cr + 1/46Mn <sub>23</sub> C <sub>6</sub> +CO+H <sub>2</sub> O	(22)
$1/2Fe_2MnO_4+2H_2=Fe+1/2Mn+2H_2O$	(23)
1/2Fe <sub>2</sub> MnO <sub>4</sub> + 1/14Cr <sub>7</sub> C <sub>3</sub> + 11/7CO = Fe + 1/2Mn + 25/14CO <sub>2</sub> + 1/2Cr	(24)

The results of  $\Delta G^0$  calculating for the reactions 13 ... 22 are shown in Fig. 7.



Fig.7. Gibbs energy of the reactions of carbon-thermal (a) and complex (b) reduction of manganese ferrite with carbide  $Cr_7C_3$  participation: 1-10 - reactions 13 ... 22 respectively.

In the above variants, the formation of joint carbides is not taken into account as for such compounds there is no thermo-chemical data. Varying the composition of the resulting carbide phase to some extent allows us to estimate the thermodynamic probability of the formation of such carbides. The iron formed in the first stage of reduction can: replace chromium atoms in the structure of carbide Cr<sub>23</sub>C<sub>6</sub> (or Cr<sub>7</sub>C<sub>3</sub>), forming a joint carbide; participate in formation of Fe<sub>3</sub>C or form a phase of Fe<sub>met</sub>. The existence of Femet. is thermodynamically probable until the completion of its reduction from ferrite. The beginning of manganese reduction at a sufficient temperature significantly changes the equilibrium composition of the

gas phase - the concentration of CO increases, which can trigger the formation of Fe<sub>3</sub>C carbide by reaction  $3Fe + 2CO = Fe_3C + CO_2$ . However, the temperature conditions for the reduction of manganese oxide by chromium carbide and the equilibrium composition of the gas phase in this system will be prohibitive for the formation of iron carbide. If a previously reduced metal, for example, nickel, is present in the system, it is possible to form a solid solution of Ni-Fe, which is thermodynamically more preferable than the existence of Fe<sub>met</sub>. However, it cannot be asserted that this variant will be more probable than the formation, by means of an isomorphous substitution, of carbide -(Cr, Fe) <sub>23</sub>C<sub>6</sub>. In this case, it is necessary to consider simultaneously two parallel processes - the transformation of the carbide phase and the reduction of the oxide, as from the physical-chemical regularities of their realization, the formation of a solid product, the ligature, depends. At complex reduction, the presence of water vapor in the system also practically excludes the existence of iron carbide. So, according to our calculations at 1273 K, the equilibrium concentration of H<sub>2</sub>O for Fe<sub>3</sub>C gasification, depending on the fraction  $(H_2 + H_2O)$  in the system (0.25 and 0.75), is ~ 0.283 ... 0.287%, respectively.

Comparing  $\Delta G^{0}$  calculation results for reactions 13-17 and 18-22, of the of manganese ferrite reduction, it should be noted a high thermodynamic preference of carbothermal process, which is related to physical-chemical features of the gasification unit. Partial replacement of carbide carbon by hydrogen, reaction 16 and 19, leads to a decrease in the thermodynamic probability in this temperature range. This feature is probably related to the difficulty of reduction at the last stage: MnO + H<sub>2</sub> = Mn + H<sub>2</sub>O, since MnO oxide is difficult to recover for gases in comparison with carbon.

The thermodynamic advantages of carbon dioxide gasification compared to steam are also important [6].

The change in the thermodynamic regularities in the transition from the carbon-thermal to complex reduction of Fe<sub>2</sub>MnO<sub>4</sub> is illustrated by the data presented in Figure 8. The reduction of the ferrite with CO is more preferably up to ~ 1523 K, further is more thermodynamically probable the reaction of complex reduction in the H<sub>2</sub> stream. The latter reaction prevails at T> 1443 K and above the gas-reduction reaction by hydrogen. The complex reduction of ferrite in a hydrogen stream at T> 1233 K, is more preferable than in a CO stream. These and other thermodynamic features of the manganese ferrite reduction, in part, can be explained by the complex chemical composition of the latter. Not the last role here will belong to the accepted physical-chemical model, which would indicate the sequence of transformations.



Fig.8. Temperature dependence of the Gibbs energy of the reactions: 1 - 19; 2 - 23; 3 - 18; 4 - 13; 5 - 24.

Based on the foregoing, it can be stated that it is thermodynamically possible to obtain a complex composition sponge ligature based on chromium by carbon-thermal or complex reduction of a mixture of oxides (ferrites) involving chromium carbides.

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